

## Objectives

$Mn_5Ge_3(001)/Ge(111)$  epitaxial films: a potential source of polarized carriers directly into Ge.  
 Interesting features: high spin polarization (42%), easy fabrication, high Curie temperature ( $T_C = 296$  K), and full compatibility with Si and Ge based technology

Doping with carbon:

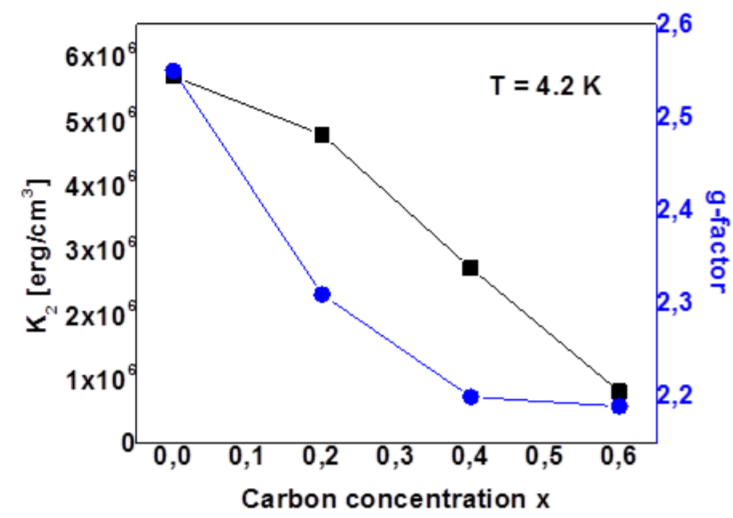
- Increased Curie temperature from 296 K up to 430 K
- Decreased uniaxial magnetocrystalline anisotropy, evidenced by FMR

Kittel's equations:

$$\left(\frac{\omega}{\gamma}\right)^2 = H_{\parallel} \left( H_{\parallel} + 4\pi M - \frac{2K_u}{M} \right)$$

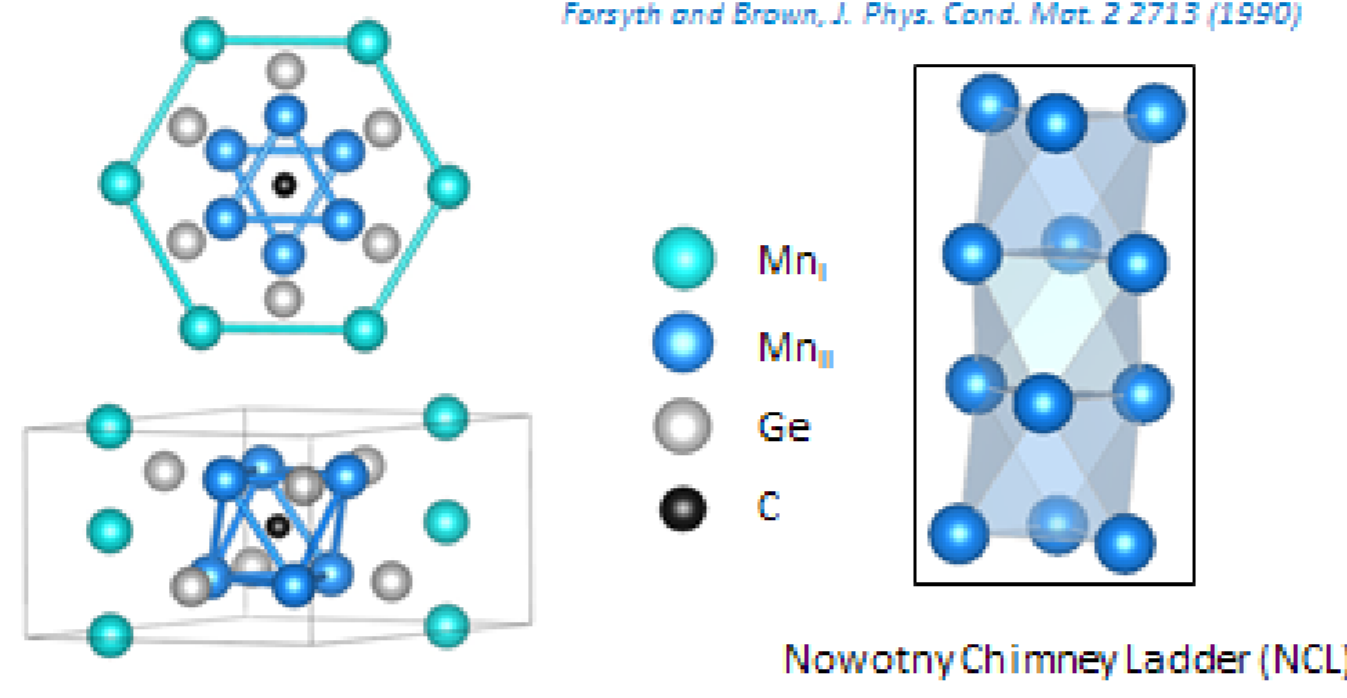
$$\frac{\omega}{\gamma} = H_{\perp} - 4\pi M + \frac{2K_u}{M}$$

$$\gamma = \frac{g\mu_B}{\hbar}$$



## $Mn_5Ge_3$ - crystal structure

- Nowotny phase, hexagonal structure - space group  $P6_3/mcm$
- Mn positions:  $Mn_I$  (4d):  $\mu = 1.94 \mu_B$  and  $Mn_{II}$  (6g):  $\mu = 3.34 \mu_B$



Nowotny Chimney Ladder (NCL)

$Mn_5Ge_3C_x$ : Carbon enters into 2(b) voids in the center of  $Mn_{II}$  octahedron in a highly ordered manner, maximum uptake:  $x=0.5$  every second 2(b) void occupied

R. Kalvig et al., Phys. Rev. B 105 094405 (2022)

Aim of this study: to understand the origin of magnetocrystalline anisotropy in the entire concentration range

## Methods

### Experimental tool: $^{55}Mn$ NMR

$$\nu_{NMR} = \gamma B_{eff} \approx B_{hf} \approx B_{hf}^{CF} + B_{hf}^{orb} = -A_{CF} \mu_{loc}^S + A_{orb} \mu_{loc}^L$$

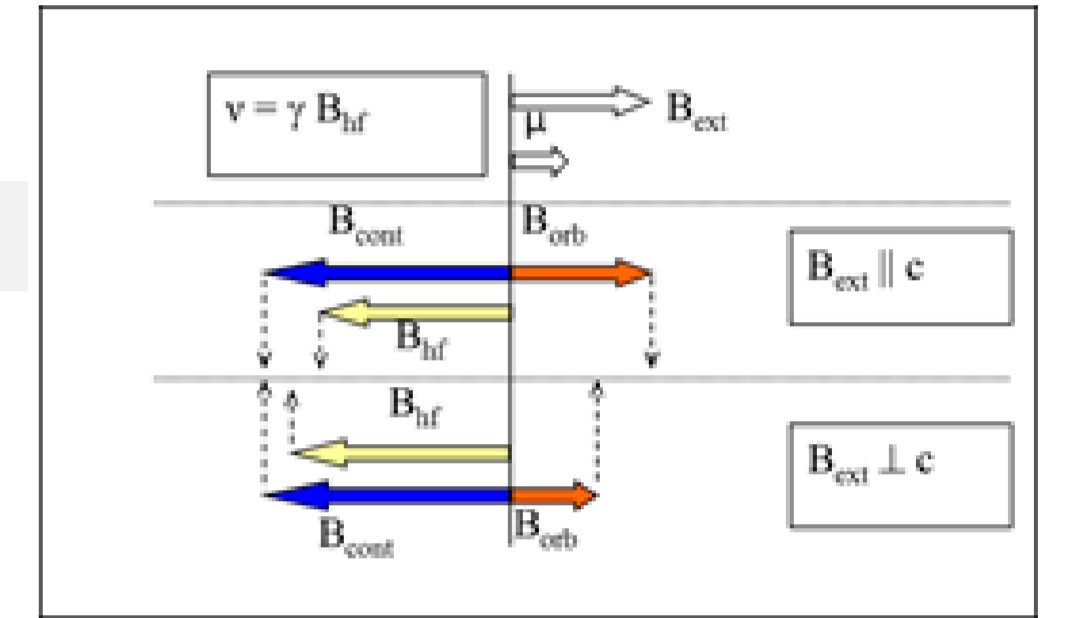
isotropic anisotropic

From calculations of hyperfine fields in  $Mn_5Ge_3$

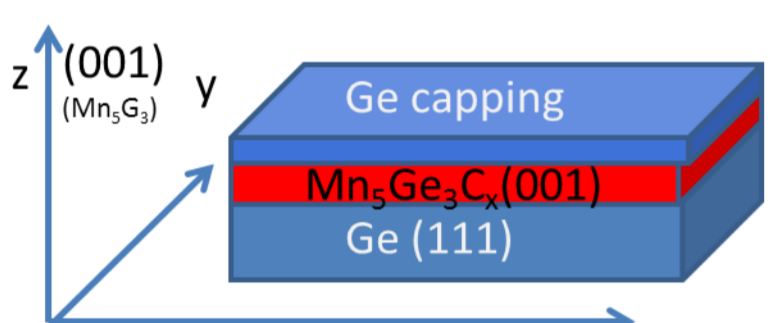
S. Piccozzi, A. Continenza, and A. J. Freeman, Phys. Rev. B 70, 235205 (2004)

$$A_{CF} = 12 T/\mu_B \quad A_{orb} = 27 T/\mu_B$$

$$^{55}Mn: \gamma = 10.553 \frac{MHz}{T}$$

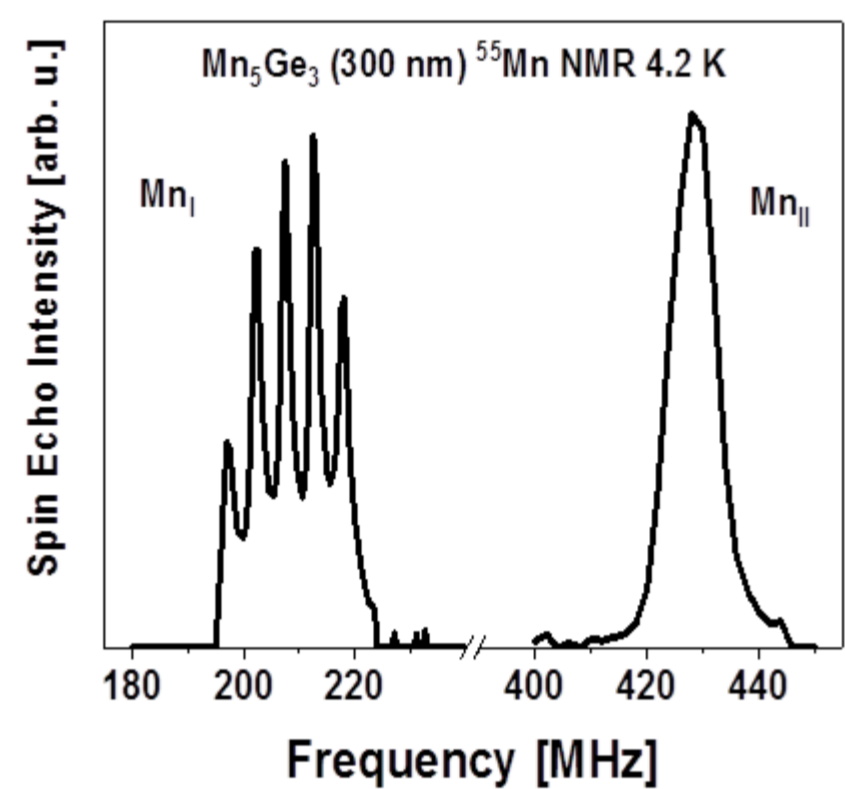
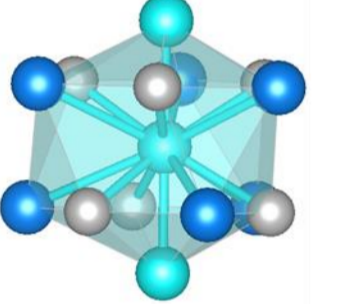


## Results: $^{55}Mn$ NMR in the undoped $Mn_5Ge_3$ film (300 nm)

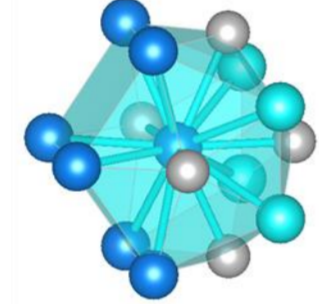


### Zero-Field $^{55}Mn$ NMR

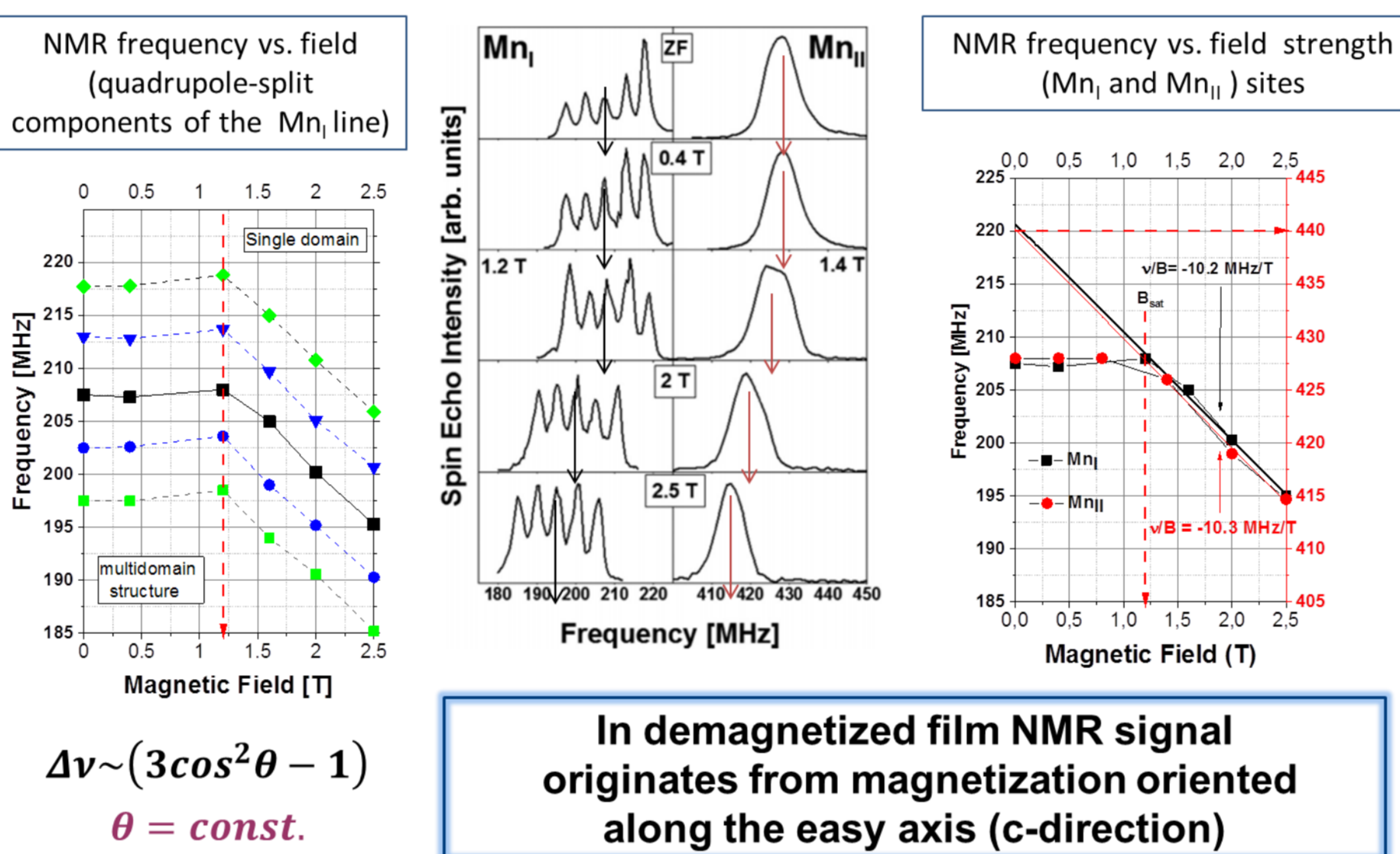
$Mn_I$  in (4d):  
 $\mu_{Mn_I} = 1.94 \mu_B$   
 $\nu_{Mn_I} = 207.5 MHz$   
 $B_{Mn_I} = 19.67 T$



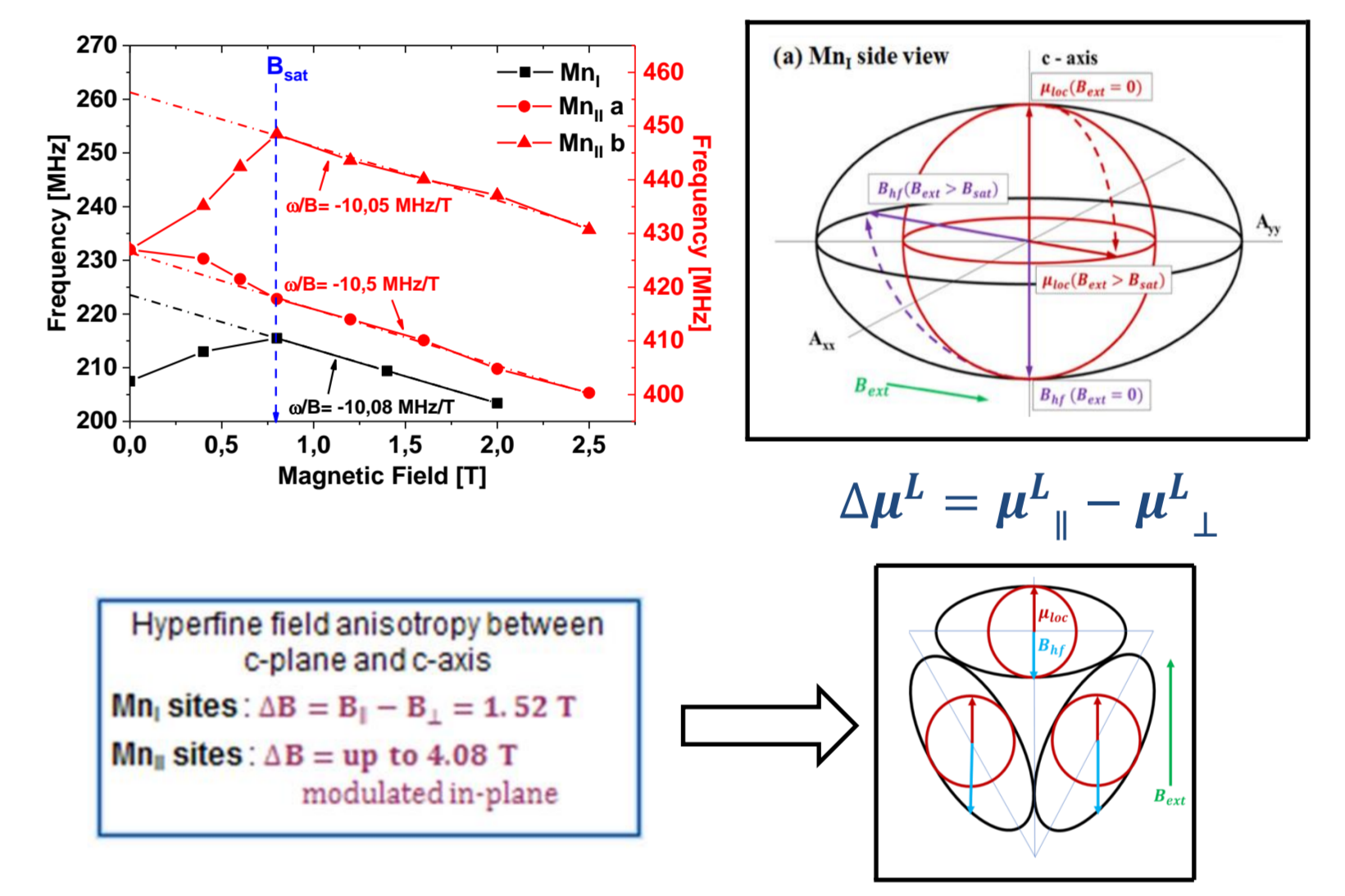
$Mn_{II}$  in (6g):  
 $\mu_{Mn_{II}} = 3.34 \mu_B$   
 $\nu_{Mn_{II}} = 428 MHz$   
 $B_{Mn_{II}} = 40.56 T$



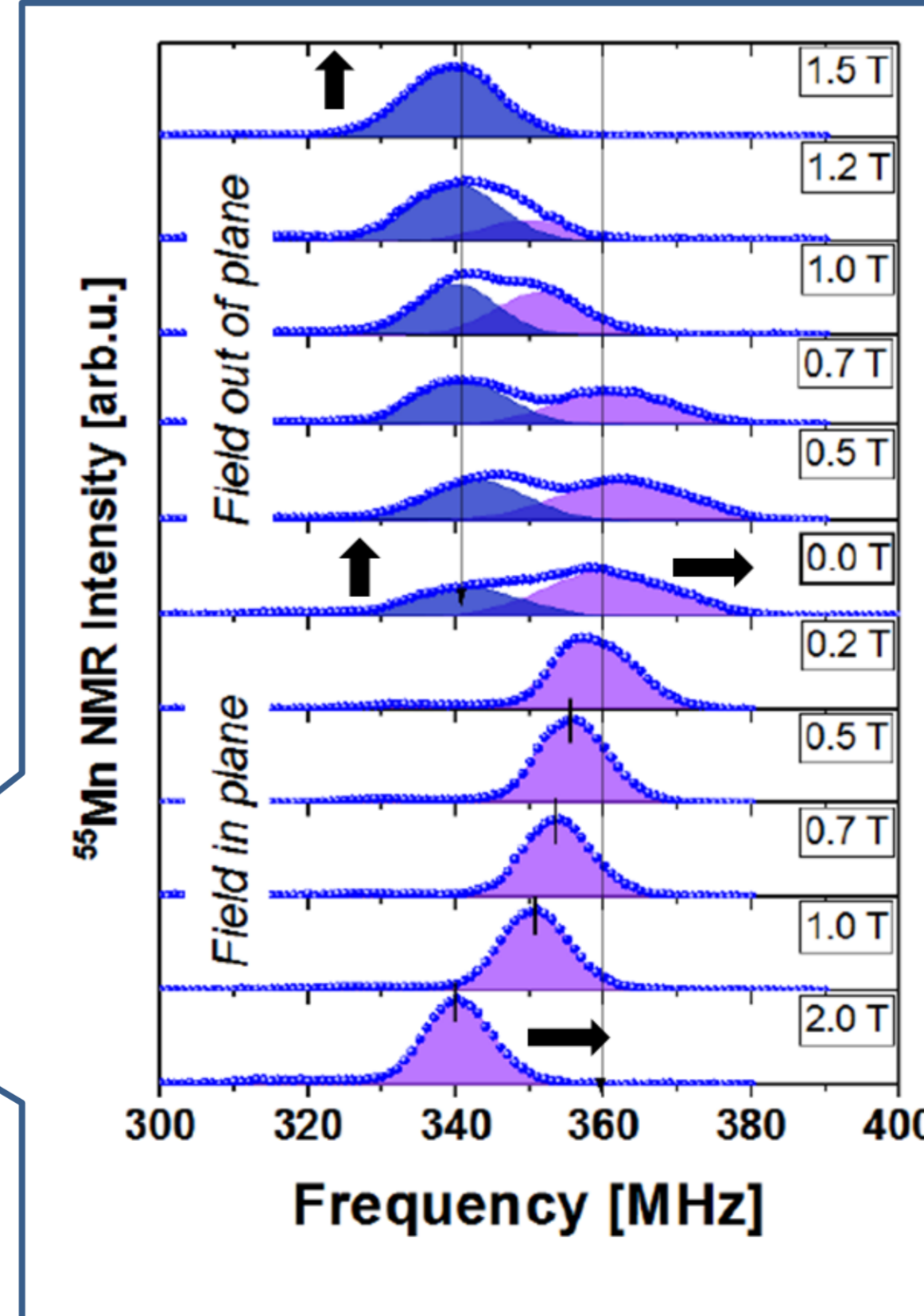
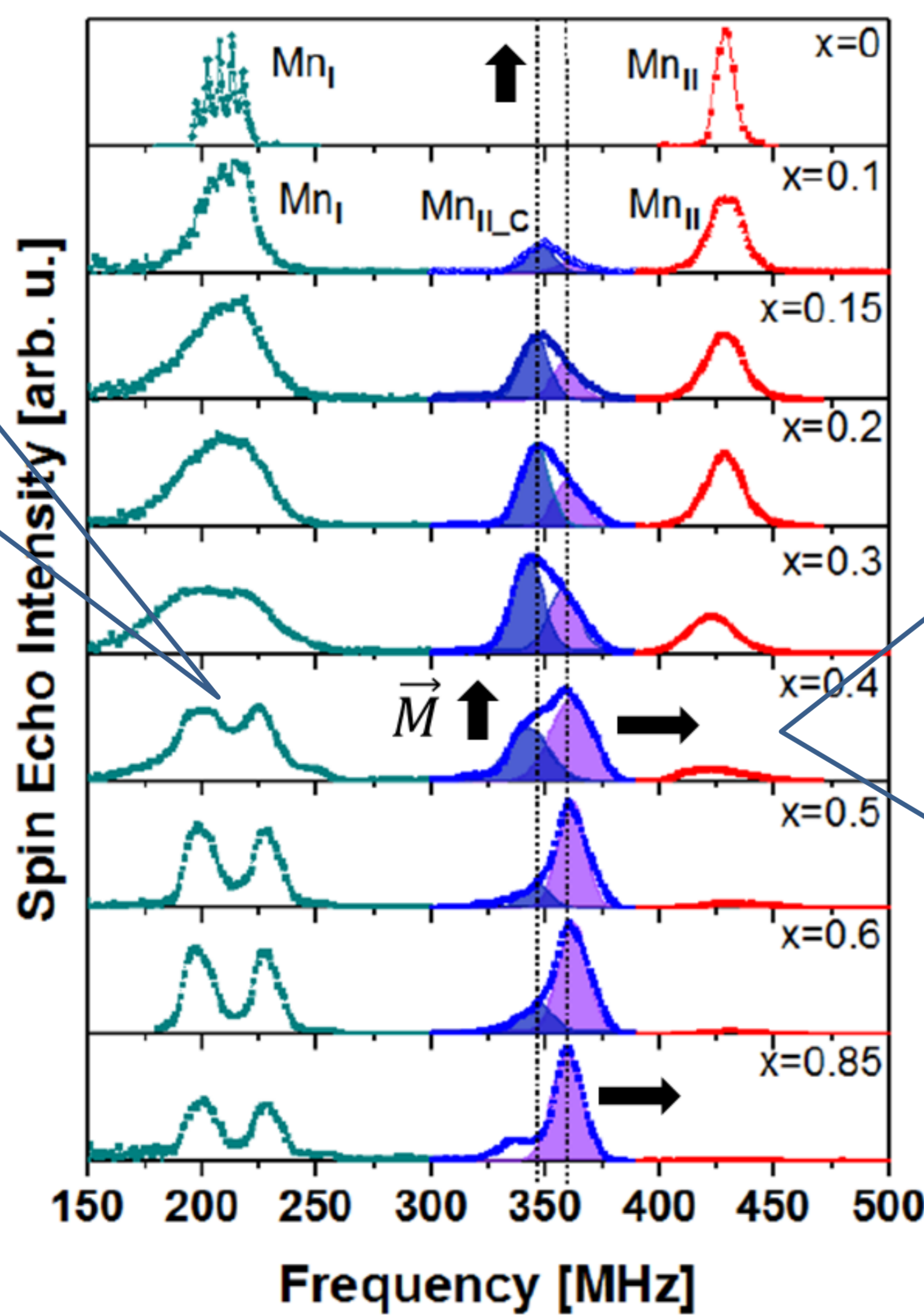
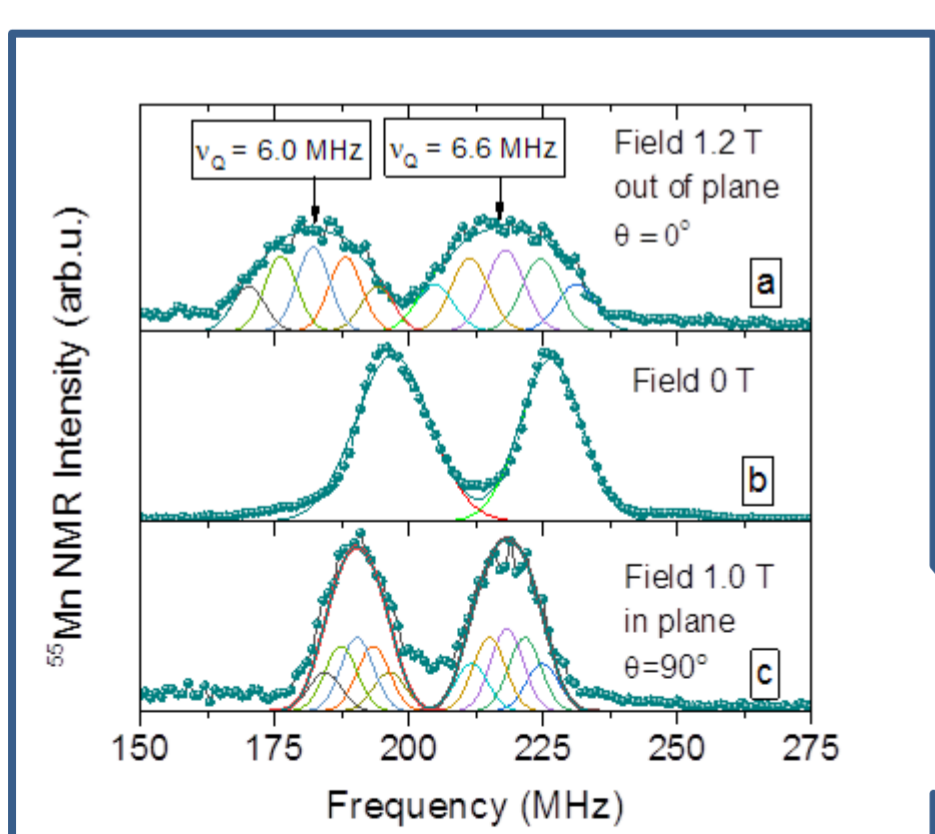
### External magnetic field perpendicular to the film



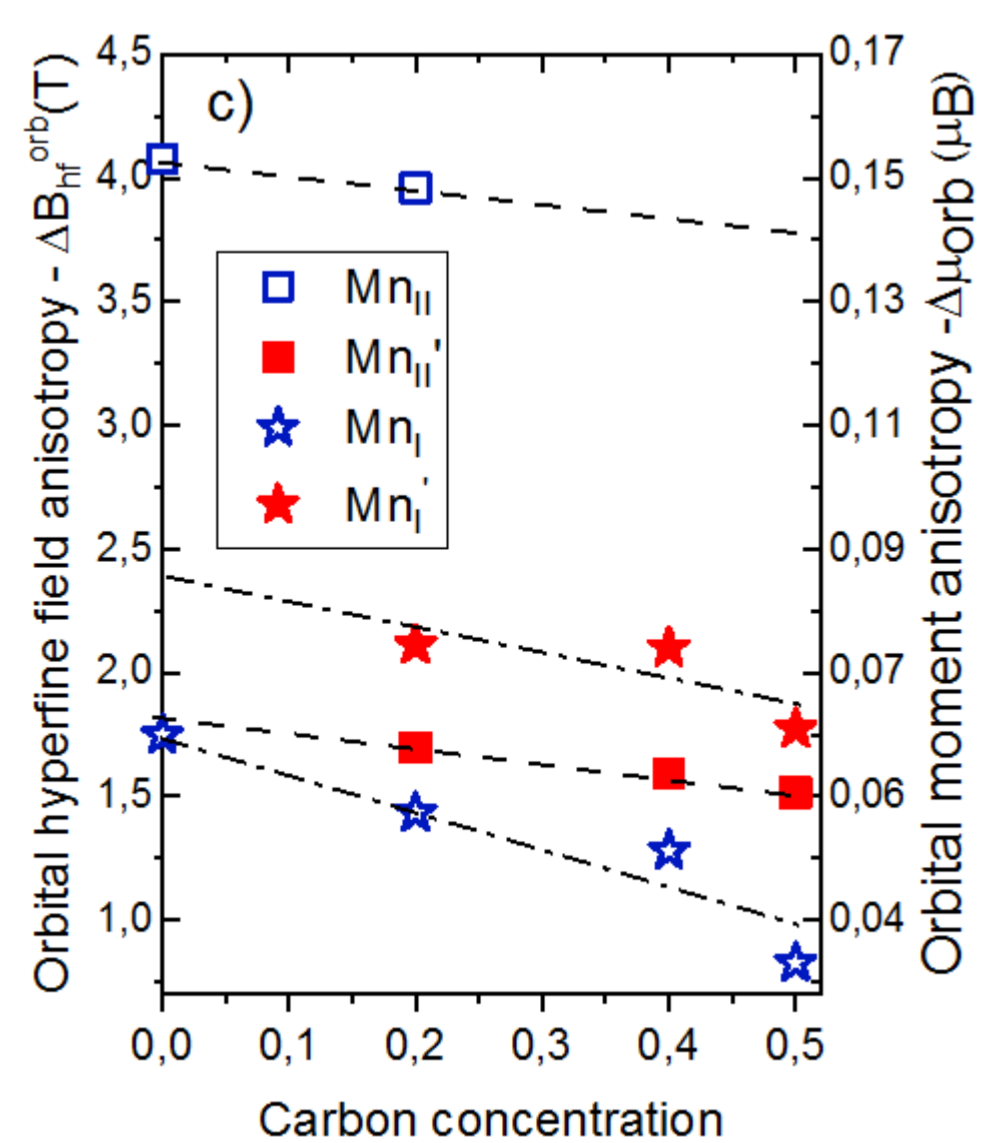
### External magnetic field in-plane of the film



## Results: $^{55}Mn$ NMR in $Mn_5Ge_3C_x$ films ( $t = 30$ nm; $0 \leq x \leq 0.85$ )



$Mn_{II}$  atoms with a carbon neighbour new spectrum component at 344 MHz due to co-existing signals from in-plane and out-of-plane orientation



$x \geq 0.5$ : No changes in the NMR spectrum

## Conclusions

- Strong anisotropy of  $^{55}Mn$  hyperfine fields between the hexagonal c-axis and the c-plane in  $Mn_I$  and  $Mn_{II}$  lattice sites within pristine  $Mn_5Ge_3$ .  $Mn_{II}$  site: an additional modulation of hyperfine field in the c-plane.
- Isotropic Fermi contact term constitutes the main component of hyperfine fields, but a significant anisotropic contribution due to the anisotropy of orbital moment  $\mu_L$  is evidenced by the NMR data.
- Carbon dopant occupies selectively interstitial 2b voids, locally changing the anisotropy of orbital contribution to the hyperfine field on those  $Mn_I$  atoms that form a host octahedron. The modulation of anisotropy within the c-plane is lifted.  $Mn_I$  sites, more distant from the dopant location are less affected.
- A linear drop of hyperfine field anisotropy upon carbon doping corresponds to the change of bulk uniaxial magnetocrystalline anisotropy evidenced by FMR, and reflects the increasing number of  $Mn_{II}$  atoms affected by their carbon neighbors.
- The above observations confirm that the magnetocrystalline anisotropy observed in this system has a single ion origin and can be linked to the observed anisotropic orbital moment of manganese.